

Bergström–Boyce model for nonlinear finite rubber viscoelasticity: theoretical aspects and algorithmic treatment for the FE method

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Abstract One of the successful approaches to model the time-dependent behaviour of elastomers is proposed by Bergström and Boyce (JMPS 46:931–954, 1998). The model is micromechanically inspired from the relaxation of a single entangled chain in a polymer gel matrix. Although the theory of inelasticity based on multiplicative decomposition of the deformation gradient is well established, the complexity of the nonlinear evolution law as well as the nonlinear equilibrium and non-equilibrium material response necessitate a precise description of the algorithmic setting. This contribution presents for the first time a novel numerical implementation of the Bergström–Boyce model in the context of finite element analysis and elaborates theoretical aspects of the model. The thermodynamical consistency of the evolution law is proven and a parameter study with respect to material parameters has been carried out. The agreement of the model with the recent experimental data is investigated.

Keywords Finite viscoelasticity · Rubber viscoelasticity · Hysteresis · Algorithmic setting · Thermodynamical consistency

1 Introduction

Rubbery polymers exhibit both elastic and viscous material behaviour simultaneously. The non-equilibrium response of rubbery polymers attracted many scientists from applied physics, material science and engineering. Since elastomers constitute a great part in industry, the importance of precise analysis and design necessitates the incorporation of highly sophisticated material models in simulation tools.

One of the recent attempts for the description of rate-dependent material behaviour of rubbery polymers is made by Bergström and Boyce [1]. In the contribution on hand, theoretical and numerical aspects of the Bergström and Boyce model will be discussed. It will be shown that the proposed evolution law is a more general representation of the finite viscoelasticity model of Reese and Govindjee [2], with slight differences. Reese and Govindjee [2] proposed an evolution law which, when linearized around the thermodynamic equilibrium, yields the finite linear viscoelasticity model of Lubliner [3]. Ignoring the volumetric viscosity term in the finite viscoelasticity model and extending the materially linear evolution of Reese and Govindjee [2], one obtains a more general expression for finite viscoelasticity as proposed by Bergström and Boyce [1]. It will be further demonstrated that the algorithmic implementation of the model can be done similarly to the algorithm proposed by Simo and Miehe [4] for finite elastoplasticity and by Simo [5] for viscoplasticity. Integration of the evolution law will be carried out by an exponential mapping algorithm (see Weber and Anand [6], Cuitino and Ortiz [7]). Such an approach has been adopted to finite viscoelasticity by Reese and Govindjee [2]. With slight modifications, the same algorithmic structure can be restructured for the finite viscoelasticity model proposed by Reese and Govindjee [2] is constructed on the Ogden type material models (see Ogden [8]). However, the algorithm proposed in [2] covers a general framework of viscoelasticity based on a multiplicative split of deformation gradient and an evolution law as a function of elastic left Cauchy-Green tensor. The evolution law proposed has a simple linear form which includes a constant volumetric and isochoric viscosity terms. The evolution law of the *BB model* includes invariants of the inelastic right Cauchy-Green tensor, which spoils the simple representation in terms of principal stretches of

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Recently, Areias and Mats [9] developed a new finite element formulation for viscoelastic materials and employed the method for the *BB model*. The evolution law is rewritten in a Lagrangean setting in terms of rate of deformation. The residual expression stemming from the evolution law is updated for C_i and $\Delta\gamma$. We will extend the algorithmic approach adopted by Reese and Govindjee for the implementation of the *BB model* within a distinct framework of volumetric-isochoric split of the free energy density. Within this context, the generalized stresses and the moduli expressions necessary for standard and mixed Jacobian-pressure finite element formulation (Q1/P0 element) will be derived. The advantage of the proposed formulation over the formulation of Areias and Mats [9] is that inversion operations involve only 3 matrices with 3 unknowns in the eigenspace of C_i whereas the latter formulation involves lengthy operations with tensor inversions. However, this is done at the expense of accuracy of the algorithm since the invariant terms of in the evolution law are a priori assumed to be constant and updated from the history variables for a given time step.

1.1 Constitutive approaches to rubber elasticity and viscoelasticity

Many material models for the description of the elastic and viscous behaviour of rubber exist. The accuracy of a proposed viscoelasticity model depends not only on the performance of the constitutive law for the evolution of the internal variables, but also on the constitutive model describing the ground state elastic response. Recently, Marckmann and Verhulst [10] have discussed 21 rubber elasticity models and compared the performance of the formulations with respect to the capabilities to uniaxial tension, equibiaxial tension and pure shear data from Treloar [11] and Kawabata et al. [12], according to the fitting capability up to stretch level of 400% at which the isoprene rubber shows a reversible elastic behaviour without crystallization. The extended tube model of Kaliske and Heinrich [13], the micro-sphere model of Miehe et al. [14] and the Ogden model are cited as the most successful models among others. The Ogden model depends on a phenomenological description of the free energy function and the model parameters do not have any physical interpretation. The success of the extended tube model and the micro-sphere model depends on the incorporation of tube like constraints which take into account the restrictions on the motion of a free chain due to the chains in the neighbourhood or the so-called forest chains. Additional to a tube constraint part, the micro-sphere model depends on an average network stretch which is linked to the macro-continuum stretches via root averaging of the macro stretches in discrete orientation directions. The *BB model* uses the *8-chain model* of Arruda and Boyce [15] for the ground state elastic response and for the Maxwell branch of the model. The *8-chain model* provides

quite satisfactory results for uniaxial data, however, underestimates the equi-biaxial and pure shear response at moderate and large deformations due to the fixed relationship between locking stretches for different deformation modes. On the other hand, considering that the model depends only on two material parameters, it yields quite satisfactory results and the simplicity of the parameter identification makes the model popular for industrial applications.

Two molecular approaches for the description of the time-dependent behaviour of polymers exist. The *transient network theory* explains the stress relaxation phenomenon as a consequence of breakage and reformation of the cross-links constantly. The theory is firstly proposed by Green and Tobolsky [16] and further developed by Lodge [17], Phan-Thien [18] and Tanaka and Edwards [19, 20]. *Reptation-type tube models* are developed for the definition of the motion of a single chain in a polymer gel. The constraints on the free motion of a single chain are qualitatively modeled as a tube-like constraint and the motion of the chain is described as a combination of Brownian motion within and reptational motion along the tube. The model is proposed by De Gennes [21] and Doi and Edwards [22]. The reptational motion is successfully adopted to finite viscoelasticity by Bergström and Boyce [1]. Recently, Miehe and Göktepeli [23]

proposed a finite viscoelasticity model depending on two micro-kinematic scalar internal variables on discrete space orientations of the micro-sphere. They propose two micro-mechanisms for the relaxation process of polymer chains: the relaxation of superimposed entanglements and the release of topological constraints is taken into account by a spectrum of nonlinear evolution laws in the logarithmic space of the discrete space orientations.

The paper is organized as follows: In Sect. 2, the basic equations governing finite viscoelasticity and the thermodynamics in terms of the concept of dissipation are introduced. Section 3 presents a complete algorithmic setting for the Bergström–Boyce model suitable for large strain finite element analysis. In Sect. 4, the efficiency of the model is validated with a benchmark example and a parameter study with respect to material parameters is carried out. Finally, concluding remarks close the paper.

2 Continuum mechanical basis of viscoelasticity

Let $\varphi : X \mapsto x$ be the deformation map at time $t \in \mathbb{R}_+$ of a body \mathcal{B} . φ maps points $X \in \mathcal{B}$ of the reference configuration $\mathcal{B} \subset \mathbb{R}^3$ onto points $x = \varphi(X; t) \in \mathcal{S}$ of the current configuration $\mathcal{S} \subset \mathbb{R}^3$. Let $F := \nabla \varphi(X; t)$ denote the deformation gradient with the determinant $J := \det F > 0$. The material time derivative of the deformation gradient is $\dot{F} = IF$ where $I := \nabla v$ is the spatial velocity gradient. The boundary-value problem for a general inelastic body is governed by the balance of momentum

$$\rho_0 \ddot{\phi} = \text{div}[\tau/J] + \gamma \tag{1}$$

along with prescribed displacement boundary conditions $\bar{\phi}(X; t)$ on $\partial \mathcal{B}_t$ and traction boundary conditions \bar{t} with outward normal n . ρ_0 is the density and γ is the prescribed body force with respect to unit volume of the current configuration. Let furthermore $g = \delta_{ab}$ denote the covariant Cartesian metric tensor or the so-called Kronecker symbol in the current configuration. Due to nearly incompressible behaviour of elastomers, we will consider the split of the elastic response into the volumetric and isochoric parts and the viscous effects are assumed to be purely isochoric

$$\bar{F} := J^{-1/3} F. \tag{2}$$

The unimodular part of the deformation gradient is further split into elastic and inelastic contributions

$$\bar{F} = F^e F^i. \tag{3}$$

The Kirchhoff stress τ is a function of the deformation gradient. A decoupled volumetric-isochoric structure of finite viscoelasticity is obtained by the special form of the Helmholtz free energy function for a unit reference volume

$$\Psi = U(J) + \bar{\Psi}(g, F^i, F^e). \tag{4}$$

The proposed form of stored energy yields an additive decomposition of the Kirchhoff stress into spherical and deviatoric contributions

$$\tau = \tau_{\text{vol}} + \tau_{\text{iso}} \quad \text{where} \tag{5}$$

$$\tau_{\text{vol}} = p g^{-1} \quad \text{and} \quad \tau_{\text{iso}} = \mathbb{P} : \bar{\tau}$$

using $\bar{\tau} := 2\partial_g \bar{\Psi}(g, F^i, F^e)$, $p := JU'(J)$ and the fourth-order deviatoric projection tensor $\mathbb{P}_{cd}^{ab} = [\delta_c^a \delta_d^b + \delta_d^a \delta_c^b]/2 - \delta^{ab} \delta_{cd}/3$. The isochoric part of the free energy is further decomposed into elastic and viscous contributions

$$\bar{\Psi} = \bar{\Psi}^e(\bar{C}) + \bar{\Psi}^v(C_e). \tag{6}$$

$\bar{C} = \bar{F}^T g \bar{F}$ and $C_e = F^{eT} g F^e$ are the unimodular part of the right Cauchy-Green tensor and the elastic right Cauchy-Green tensor, respectively. The above definition induces a further split of $\bar{\tau}$ defined in (5)₂

$$\bar{\tau} := \bar{\tau}^e + \bar{\tau}^v \quad \text{with} \tag{7}$$

$$\bar{\tau}^e := 2\partial_g \bar{\Psi}^e(\bar{C}) \quad \text{and} \quad \bar{\tau}^v := 2\partial_g \bar{\Psi}^v(C_e).$$

2.1 Free energy function

The volumetric part of the free energy is described by the simple form

$$U(J) = \kappa(J - \ln J - 1). \tag{8}$$

¹ Although the elastic right Cauchy-Green tensor is unimodular, the overline symbol ($\bar{\cdot}$) is dropped from the expression in order to simplify the notation. The same applies to inelastic right Cauchy-Green tensor C_i and elastic left Cauchy-Green tensor

κ is the bulk modulus and will be taken large enough in order to enforce the incompressibility. For the isochoric response of the material, the free energy functions $\bar{\Psi}^e$ and $\bar{\Psi}^v$ are introduced as

$$\bar{\Psi}^e(\lambda_r) := \mu N \left(\lambda_r \mathcal{L}^{-1}(\lambda_r) + \ln \frac{\mathcal{L}^{-1}(\lambda_r)}{\sinh \mathcal{L}^{-1}(\lambda_r)} \right) \tag{9}$$

$$\bar{\Psi}^v(\lambda_r^e) := \mu_v N_v \left(\lambda_r^e \mathcal{L}^{-1}(\lambda_r^e) + \ln \frac{\mathcal{L}^{-1}(\lambda_r^e)}{\sinh \mathcal{L}^{-1}(\lambda_r^e)} \right). \tag{10}$$

$\lambda_r = \sqrt{I_1/3N}$ and $\lambda_r^e = \sqrt{I_1^e/3N_v}$ denote the relative network stretches for the equilibrium and non-equilibrium deformations $I_1 := \text{tr } \bar{C}$ and $I_1^e := \text{tr } C_e$ are the first invariants of the total and elastic right Cauchy-Green tensors, respectively. μ and μ_v are the shear modulus of the elastic chain network and superimposed free chain network, whereas N and N_v are the segment numbers of the elastic and superimposed networks, respectively. The idealized model representation of a rubber network by the 8-chain model is depicted in Fig. 1.

2.2 Thermodynamical consistency

The second axiom of thermodynamics restricts the proposed model by the so-called dissipation inequality

$$\mathcal{D} := \mathcal{P} - \dot{\Psi} \geq 0 \quad \text{with} \quad \mathcal{P} := S : \frac{1}{2} \dot{C} = \tau : d. \tag{11}$$

$d := (g l + l^T g)/2$ is the symmetric Eulerian rate of the deformation tensor. Following Miehe [24], the dissipation inequality can be split into spherical and unimodular parts. Since the volumetric and isochoric responses are a priori assumed to be independent, each should satisfy the inequality separately, leading to

$$\mathcal{D} = \mathcal{D}_{\text{vol}} + \mathcal{D}_{\text{iso}} \geq 0, \tag{12}$$

$$\mathcal{D}_{\text{vol}} := \tau_{\text{vol}} : d^{\text{vol}} - \dot{U}(J) \geq 0 \quad \text{and} \tag{13}$$

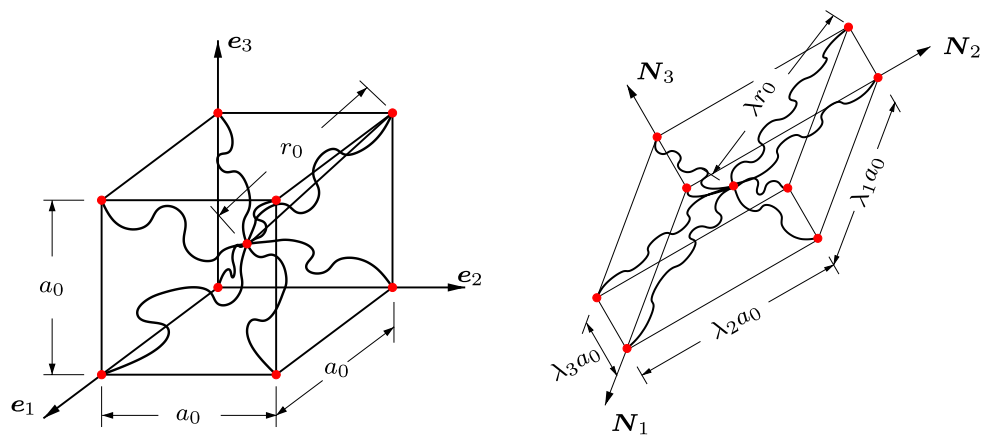
$$\mathcal{D}_{\text{iso}} := \tau_{\text{iso}} : d^{\text{iso}} - \dot{\bar{\Psi}} \geq 0.$$

For the derivation of the additive split $d = d^{\text{vol}} + d^{\text{iso}}$ and the additive split of the dissipation inequality, the reader is referred to Appendix A. No volumetric viscous effects are assumed and \mathcal{D}_{vol} vanishes for $p = JU'(J)$. Taking the material time derivative of the isochoric part of the free energy function (6), we obtain the following rate expression

$$\dot{\bar{\Psi}} = 2\partial_{\bar{C}} \bar{\Psi}^e : \frac{1}{2} \dot{\bar{C}} + 2\partial_{C_e} \bar{\Psi}^v : \frac{1}{2} \dot{C}_e. \tag{14}$$

² For convenience, the reference density appearing before $\dot{\Psi}$ is dropped from (11)₁ and the free energy density is defined per unit reference volume instead of unit mass.

Fig. 1 8-chain model representation of rubber network in the undeformed and deformed con guration



Furthermore, the covariant rate of deformation tensor $\dot{l}^{iso} = [l^{isoT} \dot{g} + \dot{g} l^{iso}]/2$ in the current con guration can be defined. Hence, $\dot{C}/2 = \varphi^*(\dot{d}) = \bar{F}^T \dot{d}^{iso} \bar{F}$, where $\varphi^*(\cdot)^b = \bar{F}^T (\cdot) \bar{F}$ is the volume preserving covariant pull-back operation. The total and elastic part of the isochoric spatial velocity gradient are related as

$$\dot{l}^{iso} = l_e + F^e l_i F^{e-1} = l_e + \tilde{l}_i. \tag{15}$$

\tilde{l}_i is the inelastic velocity gradient in the spatial con guration. This yields

$$d^{iso} := \text{sym}(g l^{iso}) = d_e + \tilde{d}_i. \tag{16}$$

$\dot{C}_e/2 = \varphi^*(\dot{d}_e)^b = F^{eT} \dot{d}_e F^e$ can also be defined with $\varphi^*(\cdot)^b$ denoting the covariant pull-back operation to the intermediate con guration. Incorporating (16) along with the expressions for \dot{C} and \dot{C}_e into (14) leads to

$$\begin{aligned} \dot{\Psi} &= 2\partial_{\dot{C}} \bar{\Psi}^e : [\bar{F}^T d^{iso} \bar{F}] + 2\partial_{\dot{C}_e} \bar{\Psi}^v : [F^{eT} \dot{d}_e F^e] \\ &= \bar{F} [2\partial_{\dot{C}} \bar{\Psi}^e] \bar{F}^T : d^{iso} + F^e [2\partial_{\dot{C}_e} \bar{\Psi}^v] F^{eT} : \dot{d}_e \\ &= \bar{\tau}^e : d^{iso} + \bar{\tau}^v : \dot{d}_e \\ &\stackrel{(16)}{=} \bar{\tau}^e : d^{iso} + \bar{\tau}^v : [d^{iso} - \tilde{d}_i] \\ &= [\bar{\tau}^e + \bar{\tau}^v] : d^{iso} - \bar{\tau}^v : \tilde{d}_i. \end{aligned} \tag{17}$$

Substituting (17) into (13) gives the dissipation inequality

$$\mathcal{D}_{iso} = \bar{\tau}^v : \tilde{d}_i \geq 0. \tag{18}$$

Since \tilde{d}_i is a deviatoric tensor, (18) can alternatively be written as

$$\mathcal{D}_{iso} = \tau_{iso}^v : \tilde{d}_i \geq 0. \tag{19}$$

2.3 Evolution law

Consistent with (9), Bergström and Boyce [1] proposed an evolution for the inelastic rate of deformation tensor in the current con guration

$$\tilde{d}_i := \dot{\gamma} N, \tag{20}$$

where $N = \frac{\tau_{iso}^v}{\|\tau_{iso}^v\|}$ with $\tau_{iso}^v := P : \bar{\tau}^v$ and $\|\tau_{iso}^v\| := \sqrt{\tau_{iso}^v : \tau_{iso}^v}$. (21)

$\dot{\gamma}$ denotes the effective creep rate, and the thermodynamical consistency (19) is satisfied for $\dot{\gamma} \geq 0$. The BB model, constitutively describes the $\dot{\gamma}$. If a polymer network is deformed at high enough rate, both the cross-linked strong network and the superimposed entanglements deform af rely. Hence, the free dangling chains contribute to the deformation resistance. If the network is kept stretched, the dangling ends of the free chains will start moving in a combination of Brownian and reptational motion as described by Doi and Edwards [23]. A descriptive picture showing a superimposed free chain on a strong elastic network is given in Fig. 2. According to the tube model of Doi and Edwards [22], a function $\phi(t)$ describes the motion of a free chain

$$\phi(t) \approx \begin{cases} t^{1/2} & t \leq \tau_e \\ t^{1/4} & \tau_e \leq t \leq \tau_R \\ t^{1/2} & \tau_R \leq t \leq \tau_d \\ t & \tau_d \leq t \end{cases}, \tag{22}$$

where τ_e denotes the time at which the tube constraint is first felt, τ_R is the Rouse relaxation time, and τ_d denotes the tube disengagement time. The effective length of the free chain can be written as

$$l(t) := l_0 + c_1 \sqrt{\phi(t)}. \tag{23}$$

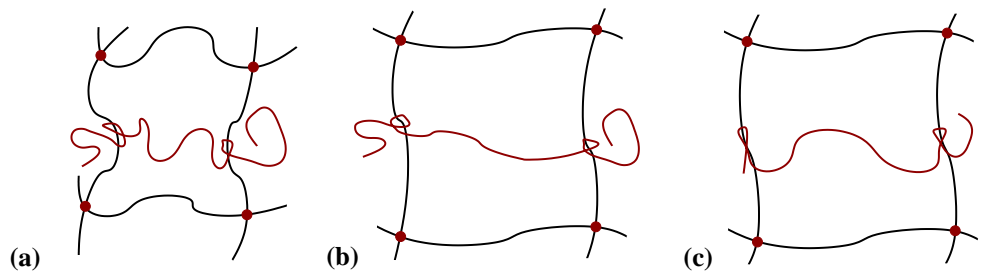
Bergström and Boyce [1] proposed the following expression for the inelastic network stretch

$$\lambda^i(t) := \frac{l(t)}{l_0} = 1 + c_2 t^{c_3}, \tag{24}$$

where $c_2 > 0$ and $c_3 \in [0.125, 0.5]$. The time derivative of (23) yields

$$\dot{\lambda}^i(t) = c_2 c_3 t^{c_3-1}. \tag{25}$$

Fig. 2 Micromechanical representation of superimposed free chains in **a** undeformed, **b** deformed state after instantaneous loading and **c** deformed and relaxed state



Incorporation of $\bar{\tau}$ in (24) into (25) gives

$$\dot{\lambda}^i(t) = c_4[\lambda^i - 1]^{c_\tau}, \tag{26}$$

where $c_\tau \in [-7, -1]$. Inserting the expression (26) into (20) yields

$$\dot{\gamma} := \dot{\gamma}_0[\lambda_{\text{chain}}^i - 1]^c \left(\frac{\tau_v}{\bar{\tau}}\right)^m. \tag{27}$$

The creep process is also assumed to be energy activated.

Therefore, the term $\left(\frac{\tau_v}{\bar{\tau}}\right)^m$ is added to the creep rate expres-

sion, where $\alpha_v := \|\tau_{\text{iso}}^v\|/\sqrt{2}$, $\lambda_{\text{chain}}^i := \sqrt{\frac{I_1^i}{3}}$ is the inelastic network stretch and $I_1^i := \text{tr } C_i$, $C_i := F^{iT} \bar{G} F^i$ is the inelastic right Cauchy-Green tensor in the reference configuration, $\bar{G} := \delta_{\bar{A}\bar{B}}$ is the metric of the intermediate configuration, $\dot{\gamma}_0$ is the reference effective creep rate, $\bar{\tau}$ is a parameter introduced for dimensional purposes and has the same units of τ_v . Power term c controls the kinetics of chain relaxation, whereas m controls the energy activated inelastic flow. In order to show the effects of the material parameters (27), a parameter study is carried out in Sect. 2.

The thermodynamical consistency of the model is satisfied for $\dot{\gamma}_0/\bar{\tau}^m > 0$ and $m > 0$. c is restricted by the tube model of Doi and Edwards (22) to be $c \in [-7, -1]$. However, to be more general, we propose $c < 0$ different from the proposition of Bergström and Boyd (14) where $c \in [-1, 0]$. By taking $c = 0$ and $m = 1$, one recovers the deviatoric part of the evolution law proposed by Reese and Govindjee (2). Thus, we can conclude that the model is a generalized version of the finite viscoelasticity model proposed by Reese and Govindjee (2).

It should be mentioned that the micro-macro transition is achieved by linking the stretch λ^i of the single chain relaxing in a polymer chain to the term α_v of the continuum. Alternative representation of the model can be obtained by substituting $\lambda^i = \lambda/\lambda^e$ in order to avoid inelastic right Cauchy-Green tensor in the evolution law, where $\lambda^e = \sqrt{\frac{\text{tr } b_e}{3}}$ and $\lambda = \sqrt{\frac{\text{tr } b}{3}}$. From this substitution, it can be concluded that the BB model intrinsically includes dependence of the viscosity on total deformation as stated in Amin et al. (25) [

3 Algorithmic setting for the constitutive model

In this section, the algorithmic setting of the model suitable for a finite element implementation will be treated. The stresses and moduli terms of the equilibrium and non-equilibrium branches will be derived. Furthermore, we propose an efficient implicit update for the principle elastic stretches of the viscous branch, conceptually following the work of Reese and Govindjee (2).

3.1 Stress and moduli terms for the elastic part

The Kirchhoff stresses and the Eulerian moduli terms corresponding to the volumetric part of the deformation can be defined as

$$\tau_{\text{vol}} := 2\partial_g U(J) \quad \text{and} \quad C_{\text{vol}} := 4\partial_g^2 U(J). \tag{28}$$

Taking the necessary derivatives, we end up with

$$\tau_{\text{vol}} = p g^{-1} \quad \text{and} \quad C_{\text{vol}} = (p + s) g^{-1} \otimes g^{-1} - 2p \mathbb{1}, \tag{29}$$

where $\mathbb{1}_{g^{-1}}^{abcd} := [\delta^{ac}\delta^{bd} + \delta^{ad}\delta^{bc}]/2$ is the fourth order identity tensor.

$$p := J U'(J) \quad \text{and} \quad s := J^2 U''(J) \tag{30}$$

are the hydrostatic Kirchhoff stresses (negative pressure) and corresponding modulus, respectively. The deviatoric stresses and moduli read

$$\tau_{\text{iso}} = \mathbb{P} : \bar{\tau} \quad \text{and} \tag{31}$$

$$C_{\text{iso}} = \mathbb{P} : [\bar{C} + \frac{2}{3}(\bar{\tau} : g) \mathbb{1} - \frac{2}{3}(\bar{\tau} \otimes g^{-1} + g^{-1} \otimes \bar{\tau})] : \mathbb{P}$$

where

$$\bar{C} := \bar{C}^e + \bar{C}_{\text{algo}}^v. \tag{32}$$

$\bar{C}^e := 4\partial_{gg}^2 \bar{\Psi}^e(\lambda_r)$ denote the elasticity moduli of the equilibrium response. The algorithmic consistent moduli \bar{C}_{algo}^v for the non-equilibrium response will be explained in the next section. The stress and moduli expressions of the equilibrium response are derived from (a) as

$$\bar{\tau}^e = \frac{\mu}{3} \frac{\mathcal{L}^{-1}(\lambda_r)}{\lambda_r} \bar{b} \quad \text{and} \quad \bar{C}^e = \frac{4\mu}{9N} \frac{1}{(1 - \lambda_r^2)^2} \bar{b} \otimes \bar{b}. \tag{33}$$

$$\bar{\mathbf{b}} = \bar{\mathbf{F}}\mathbf{G}^{-1}\bar{\mathbf{F}}^T \text{ is the unimodular part of the Finger tensor, } \dot{\mathbf{b}}_e = [-2\dot{\gamma}\mathbf{N}]\mathbf{b}_e^{tr}. \tag{38}$$

and \mathbf{G} , where $G_{AB} := \delta_{AB}$, is the reference metric. In the above equation, the inverse Langevin function is computed by Padé approximation $\mathcal{L}^{-1}(\lambda_r) \approx \lambda_r(3 - \lambda_r^2)/(1 - \lambda_r^2)$ proposed by Cohen [26]. It is superior to the polynomial approximations with a simple form. It shows the singular behaviour of the inverse Langevin function at $\lambda = 1$ and exhibits very good results. Diani and Gilormini [27] have reported that the error associated with the Padé approximation does not exceed 5%, where the approximation is reported to yield exact results at $\lambda_r = 0$ and around the limit $\lambda = 1$ and the deviation steadily increases from $\lambda = 0$ attaining a maximum of relative error $\approx 5\%$ around $\lambda_r = 0.8$.

3.2 Stress and moduli terms for the viscous part

From (7)₃, the viscous part of the Kirchhoff stresses can be evaluated as

$$\bar{\boldsymbol{\tau}}^v = \frac{\mu_v}{3} \frac{\mathcal{L}^{-1}(\lambda_r^e)}{\lambda_r^e} \mathbf{b}_e. \tag{34}$$

In order to compute the expression (34), one needs the current value of \mathbf{b}_e at time $t = t_{n+1}$. The computation of \mathbf{b}_e depends on the treatment of evolution law which will be explained in the next subsection.

3.2.1 Integration of the evolution law

The kernel of the subsequent treatment depends on the identity

$$-\frac{1}{2} \boldsymbol{\xi}_v \mathbf{b}_e \cdot \mathbf{b}_e^{-1} = \tilde{\mathbf{d}}_i. \tag{35}$$

The integration of the evolution law is based on an operator split of the material time derivative of \mathbf{b}_e into an elastic predictor (E) and an inelastic corrector step (I)

$$\dot{\mathbf{b}}_e := \underbrace{\dot{\mathbf{l}}^{\text{iso}} \mathbf{b}_e + \mathbf{b}_e \dot{\mathbf{l}}^{\text{iso}T}}_E + \underbrace{\boldsymbol{\xi}_v \mathbf{b}_e}_I. \tag{36}$$

$\boldsymbol{\xi}_v \mathbf{b}_e := \bar{\mathbf{F}}\dot{\mathbf{C}}_i^{-1}\bar{\mathbf{F}}^T$ is a unimodular operator similar to the Lie time derivative of the elastic Finger tensor.

During the *elastic trial step*, $\dot{\mathbf{C}}_i^{-1}$ is equal to zero. Therefore,

$$(\mathbf{C}_i^{-1})_{tr} = (\mathbf{C}_i^{-1})_{t_n} \rightarrow \mathbf{b}_e^{tr} = \bar{\mathbf{F}}(\mathbf{C}_i^{-1})_{t_n}\bar{\mathbf{F}}^T. \tag{37}$$

The subscript denoting the current time step, is dropped for convenience. The asset of this proposal is to circumvent the rate of inelastic metric in computation of the consistent tangent moduli.

In the inelastic corrector step $\mathbf{p} = \mathbf{l}^{\text{iso}} + \mathbf{l}^{\text{vol}}$ is set to zero leading $\boldsymbol{\xi}_v \mathbf{b}_e = \dot{\mathbf{b}}_e$. Consequently, we have the following for the evolution law

The equation can be solved by the so-called exponential mapping. This yields the expression

$$\mathbf{b}_e = \exp \left[-2 \int_{t_n}^{t_{n+1}} \dot{\gamma} \mathbf{N} dt \right] \mathbf{b}_e^{tr}. \tag{39}$$

Approximating the integral equation, we obtain

$$\mathbf{b}_e \approx \exp[-2\dot{\gamma}\mathbf{N}\Delta t] \mathbf{b}_e^{tr}. \tag{40}$$

Due to isotropy, $\boldsymbol{\tau}^v$ and consequently \mathbf{N} commute with \mathbf{b}_e and also with \mathbf{b}_e^{tr} . Hence, (39) can be written in the principal stretch directions

$$\mathbf{b}_e^{e2} \approx \exp \left[-\frac{2}{\|\text{dev} \boldsymbol{\tau}^v\|} \Delta t \dot{\gamma} \text{dev} \boldsymbol{\tau}_a \right] \lambda_a^{e2}. \tag{41}$$

Introducing elastic principal logarithmic stretches $\varepsilon_a = \ln \lambda_a^{e2}$, and taking the logarithm of (41), we obtain

$$\varepsilon_a \approx -\frac{\Delta t}{\sqrt{2}} \frac{\dot{\gamma}}{\tau_v} \text{dev} \boldsymbol{\tau}_a + \varepsilon_a^{tr}. \tag{42}$$

(42) constitutes a nonlinear relation since ε_a and $\dot{\gamma}$ are also a function of principal stretches. Hence, the residual expression for iterative solution will be defined

$$\mathbf{r}_a := \varepsilon_a + \frac{\Delta t}{\sqrt{2}} \frac{\dot{\gamma}}{\tau_v} \text{dev} \boldsymbol{\tau}_a - \varepsilon_a^{tr} = 0. \tag{43}$$

The steps of the local Newton iteration are summarized in Table 1. The derivation of the local tangent \mathbf{K} is shown in Appendix B. During the computation of the residual tangent \mathbf{K} , inelastic network stretch is frozen such that $\lambda_{\text{chain}}^i = \lambda_{\text{chain}^i|t_n}^i$ in order to avoid the partial derivative $\frac{\partial \lambda_{\text{chain}}^i}{\partial \varepsilon_b}$. This is due to the fact that the $\boldsymbol{\varepsilon}^i$ and \mathbf{b}_e do not lie in the same eigenspace. This assumption disturbs the unconditional stability of the implicit update scheme used. However, as shown in Sect. 4.5, the algorithm is robust with no indication stability problems. The computational time is improved significantly, since lengthy tensorial operations are avoided in the internal iteration step.

3.2.2 Algorithmic moduli for the viscous part

In this section, we will set up a closed-form expression for the so-called consistent tangent moduli for the viscous part. The spectral decomposition of the trial elastic deformation yields

$$\mathbf{F}_e^{tr} = \sum_{a=1}^3 \lambda_a^{e2} \mathbf{n}_a \otimes \mathbf{n}_a. \tag{44}$$

Table 1 Steps of local Newton iteration

1. Set initial values	$k = 0, \varepsilon_a^k = \varepsilon_a^{tr}$
DO	
2. Residual equation	$r_a := \varepsilon_a + \frac{\Delta t}{\sqrt{2}} \frac{\dot{\gamma}}{\tau_a} \text{dev} \tau_a - \varepsilon_a^{tr} = 0$
3. Linearization	$\text{Lin} r_a = r_a _{\varepsilon^k} + \frac{\partial r_a}{\partial \varepsilon_b} \Big _{\varepsilon=\varepsilon^k} \Delta \varepsilon_b^k = 0$
4. Compute	$\mathcal{K}_{ab} := \frac{\partial r_a}{\partial \varepsilon_b} \Big _{\varepsilon=\varepsilon^k}$
5. Solve	$\Delta \varepsilon_a^k = -\mathcal{K}_{ab}^{-1} r_a$
6. Update	$\varepsilon_a^{k+1} \leftarrow \varepsilon_a^k + \Delta \varepsilon_a^k$
	$k \leftarrow k + 1$
WHILE	$TOL \leq \ r_a\ $

We define the constitutive second Piola-Kirchhoff stress tensor

$$\tilde{S}^v := F_e^{tr-1} \bar{\tau}^v F_e^{tr-T}; \quad \tilde{S}^v = \sum_{a=1}^3 \frac{\tau_a}{\lambda_a^{e tr 2}} N_a \otimes N_a \quad (45)$$

in the intermediate configuration. With the definition (45)₁ at hand, the incremental rate equation can be defined

$$\Delta \tilde{S} = \bar{C}_{\text{algo}}^v : \Delta C_e^{tr} \quad (46)$$

where

$$\bar{C}_{\text{algo}}^v = 2 \partial_{C_e^{tr}} \tilde{S}. \quad (47)$$

Exploiting (47), starting from (45)₂, we end up with the moduli expression

$$\begin{aligned} \bar{C}_{\text{algo}}^v &= \sum_{a=1}^3 \sum_{b=1}^3 \frac{c_{ab} - 2\tau_a \delta_{ab}}{\lambda_a^{e tr 2} \lambda_b^{e tr 2}} N_a \otimes N_a \otimes \partial_{C_e^{tr}} \lambda_b^{e tr 2} \\ &+ \sum_{a=1}^3 \frac{2\tau_a}{\lambda_a^{e tr 2}} \partial_{C_e^{tr}} (N_a \otimes N_a). \end{aligned} \quad (48)$$

The partial derivatives are given by

$$\partial_{C_e^{tr}} \lambda_b^{e tr 2} = N_b \otimes N_b \quad (49)$$

and

$$\partial_{C_e^{tr}} (N_a \otimes N_a) = \sum_{b \neq a}^3 \frac{1}{2} \frac{1}{\lambda_a^{e tr 2} - \lambda_b^{e tr 2}} (\mathbb{G}_{ab} + \mathbb{G}_{ba}) \quad (50)$$

where $\mathbb{G}_{ab}^{IJKL} := M_a^{IK} M_b^{JL} + M_a^{IL} + M_b^{JK}$ with

$$M_a := N_a \otimes N_a \quad \text{and} \quad M_a^{IJ} := N_a^I N_a^J. \quad (51)$$

After incorporation of (49) and (50) into (48), we finally obtain the compact definition of the moduli expression

$$\begin{aligned} \bar{C}_{\text{algo}}^v &= \sum_{a=1}^3 \sum_{b=1}^3 \frac{c_{ab} - 2\tau_a \delta_{ab}}{\lambda_a^{e tr 2} \lambda_b^{e tr 2}} M_a \otimes M_b \\ &+ \sum_{a \neq b}^3 \sum_{b=1}^3 \frac{1}{2} \frac{\tilde{s}_a - \tilde{s}_b}{\lambda_a^{e tr 2} - \lambda_b^{e tr 2}} (\mathbb{G}_{ab} + \mathbb{G}_{ba}), \end{aligned} \quad (52)$$

with $\tilde{s}_a = \tau_a / \lambda_a^{e tr 2}$. In (52), the term c_{ab} is defined as

$$c_{ab} := \frac{\partial \tau_a}{\partial \varepsilon_b^{tr}} = \frac{\partial \tau_a}{\partial \varepsilon_c} \frac{\partial \varepsilon_c}{\partial \varepsilon_b^{tr}}, \quad (53)$$

and for the derivation, we recall the residual expression of the local Newton iteration. The total derivative of the residual expression with respect to trial stretches yields

$$\begin{aligned} \frac{dr_a}{d\varepsilon_b^{tr}} &= \frac{\partial r_a}{\partial \varepsilon_b^{tr}} + \frac{\partial r_a}{\partial \varepsilon_c} \frac{\partial \varepsilon_c}{\partial \varepsilon_b^{tr}} = 0 \\ &= -\delta_{ab} + \mathcal{K}_{ac} \frac{\partial \varepsilon_c}{\partial \varepsilon_b^{tr}} = 0. \end{aligned} \quad (54)$$

Hence, we obtain the expression

$$\frac{\partial \varepsilon_a}{\partial \varepsilon_b^{tr}} = \mathcal{K}_{ab}^{-1}. \quad (55)$$

Inserting (55) and (B.9) into (53), we finally end up with

$$c_{ab} = \mathcal{I}_{ac} \mathcal{K}_{cb}^{-1}. \quad (56)$$

However, expression (52) is still in the constitutive intermediate configuration. By executing the push-forward operation via F_e^{tr} , the algorithmic moduli expression in the current configuration

$$\bar{C}_{\text{algo}}^v \text{ } ijkl = F_e^{tr i} F_e^{tr j} F_e^{tr k} F_e^{tr l} \bar{C}_{\text{algo}}^v \text{ } IJKL \quad (57)$$

is obtained.

The moduli expression (52) has a singularity for equal eigenvalues $\lambda_a^{e tr} = \lambda_b^{e tr}$

$$\lim_{\lambda_a^{e tr} \rightarrow \lambda_b^{e tr}} \frac{\tilde{s}_a - \tilde{s}_b}{\lambda_a^{e tr 2} - \lambda_b^{e tr 2}} = \frac{0}{0}. \quad (58)$$

Recalling L'Hospital's rule

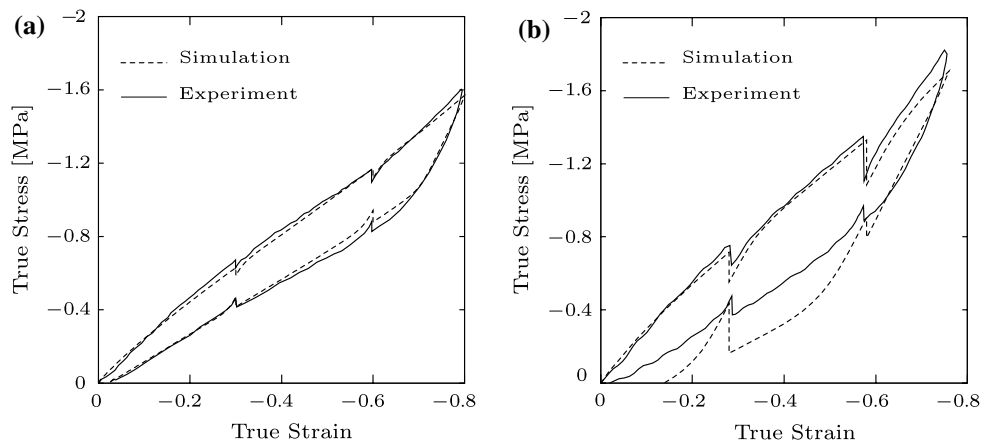
$$\lim_{x \rightarrow a} \frac{f(x)}{g(x)} = \lim_{x \rightarrow a} \frac{f'(x)}{g'(x)} \quad \text{if} \quad \lim_{x \rightarrow a} \frac{f(x)}{g(x)} = \frac{0}{0} \quad (59)$$

and applying (59) to (58) we obtain

$$\lim_{\lambda_a^{e tr} \rightarrow \lambda_b^{e tr}} \frac{\tilde{s}_a - \tilde{s}_b}{\lambda_a^{e tr 2} - \lambda_b^{e tr 2}} = \frac{1}{2} \frac{c_{aa} - 2\tau_a}{\lambda_a^{e tr 4}}. \quad (60)$$

The rest of the division by zero problems, e.g. (60) can be overcome by the perturbation of the denominator.

Fig. 3 Compression test with intermittent relaxation. **a** True strain rate $\dot{\epsilon} = -0.002 \text{ s}^{-1}$, **b** true strain rate $\dot{\epsilon} = -0.01 \text{ s}^{-1}$. Experiments are taken from Bergström and Boyce [1]



4 Numerical examples

In this section, we will assess the capability of the proposed algorithmic setting and the material model. Firstly, in mind that the evolution law is singular at the beginning of the loading process where $\dot{\lambda}_{chain} = 1$. In order to overcome this problem, a perturbation parameter ϵ is added to the term $[\lambda_{chain}^i - 1 + \epsilon]^c$ by Bergström and Boyce [30]. Due to the power term $c < 0$, the perturbation parameter should be adjusted in order to overcome the singularity at the first few steps of the loading process. After validation of the proposed algorithm, we investigate the sensitivity of the model with respect to the material parameters in the evolution law. The uniaxial compression test described is used for the investigation with $\dot{\epsilon} = -0.002$. The sensitivity of the model with respect to parameters $\dot{\gamma}_0/\hat{\tau}^m$ and m is depicted in Fig. 4. In Fig. 4a, all the other parameters are taken as given above. The smaller the value of $\dot{\gamma}_0/\hat{\tau}^m$, the more significant is the hysteresis. On the other hand, as seen in Fig. 4b, the greater the value of m , the greater is the hysteresis. In Fig. 5, the effect of the parameter $c \in [-1, 0]$ and $c \leq -1$ is studied, respectively. As the stresses (28) and moduli terms (29) drop from the formulation and the incompressibility is enforced on the element, the bulk modulus acts as a penalty parameter at element level.

yields results identical to the ones proposed by Bergström and Boyce [1] (see Fig. 3). This justifies the assumptions

made for the proposed algorithm. It should however be kept in mind that the evolution law is singular at the beginning of the loading process where $\dot{\lambda}_{chain} = 1$. In order to overcome this problem, a perturbation parameter ϵ is added to the term $[\lambda_{chain}^i - 1 + \epsilon]^c$ by Bergström and Boyce [30]. Due to the power term $c < 0$, the perturbation parameter should be adjusted in order to overcome the singularity at the first few steps of the loading process. After validation of the proposed algorithm, we investigate the sensitivity of the model with respect to the material parameters in the evolution law. The uniaxial compression test described is used for the investigation with $\dot{\epsilon} = -0.002$. The sensitivity of the model with respect to parameters $\dot{\gamma}_0/\hat{\tau}^m$ and m is depicted in Fig. 4. In Fig. 4a, all the other parameters are taken as given above. The smaller the value of $\dot{\gamma}_0/\hat{\tau}^m$, the more significant is the hysteresis. On the other hand, as seen in Fig. 4b, the greater the value of m , the greater is the hysteresis. In Fig. 5, the effect of the parameter $c \in [-1, 0]$ and $c \leq -1$ is studied, respectively. As the stresses (28) and moduli terms (29) drop from the formulation and the incompressibility is enforced on the element, the bulk modulus acts as a penalty parameter at element level.

4.1 Example 1

In the first numerical example, we consider the uniaxial compression test with intermittent relaxation time as a benchmark problem. The material is subjected to true strain rates of $\dot{\epsilon} = -0.002 \text{ s}^{-1}$ and $\dot{\epsilon} = -0.1 \text{ s}^{-1}$ with 120 s relaxation time at $\epsilon = -0.3$ and $\epsilon = -0.6$, respectively. The experimental results are taken from Bergström and Boyce [1]. The material parameters used are taken identically to the ones proposed by the authors and are $\sigma = 1,000 \text{ MPa}$, $\mu = 0.6 \text{ MPa}$, $\mu_v = 0.96 \text{ MPa}$, $N = 8$, $N_v = 8$, $\dot{\gamma}_0/\hat{\tau}^m = 7 \text{ s}^{-1} \text{ MPa}^{-m}$, $c = -1$ and $m = 4$. The proposed algorithmic implementation

4.2 Example 2

In the second example, we carry out a parametric investigation of the evolution law. The material parameters are taken as $\sigma = 1,000 \text{ MPa}$, $\mu = 0.6 \text{ MPa}$, $\mu_v = 0.96 \text{ MPa}$, $N = N_v = 8$ and $\dot{\gamma}_0/\hat{\tau}^m = 0.07 \text{ s}^{-1} \text{ MPa}^{-m}$. Taking the parameter $c = 0$ and $m = 1$, the model recovers the isochoric part of the evolution law proposed by Reese and Govindjee [2]. In Fig. 6a, normalized relaxation curves for stretch levels $\lambda = 1.1$, $\lambda = 1.5$ and $\lambda = 2.0$ are plotted against time. For $\lambda = 1.1$, the deviation is small from thermodynamical equilibrium and the relaxation curve approaches to the so-called linear

Fig. 4 Sensitivity of the model with respect to material parameter $\dot{\gamma}_0/\dot{\tau}^m$ and m : **a** $\dot{\gamma}_0/\dot{\tau}^m = [0.05, 0.1, 0.5, 1.0, 5]$, **b** $m = [2, 2.5, 3.0, 3.5, 4.0, 4.5]$

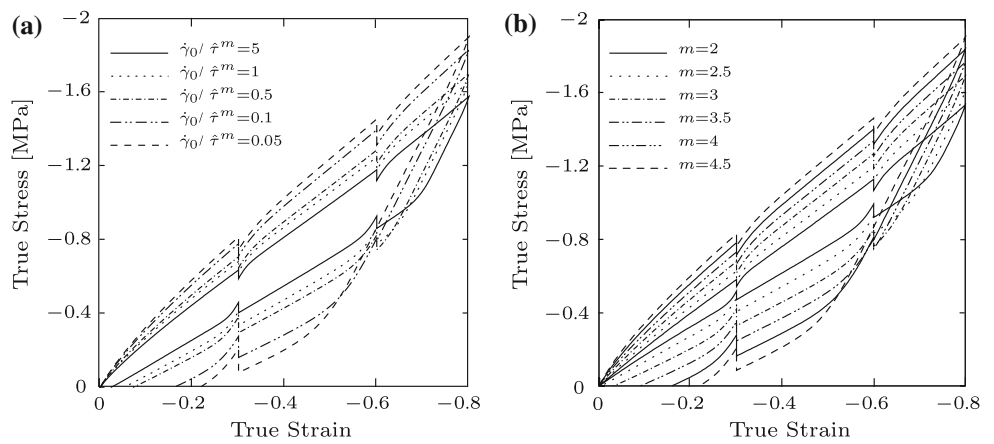


Fig. 5 Sensitivity of the model with respect to material parameter: **a** $c = [-1, -0.8, -0.6, -0.4, -0.2]$, **b** $c = [-1, -1.5, -2, -2.5]$

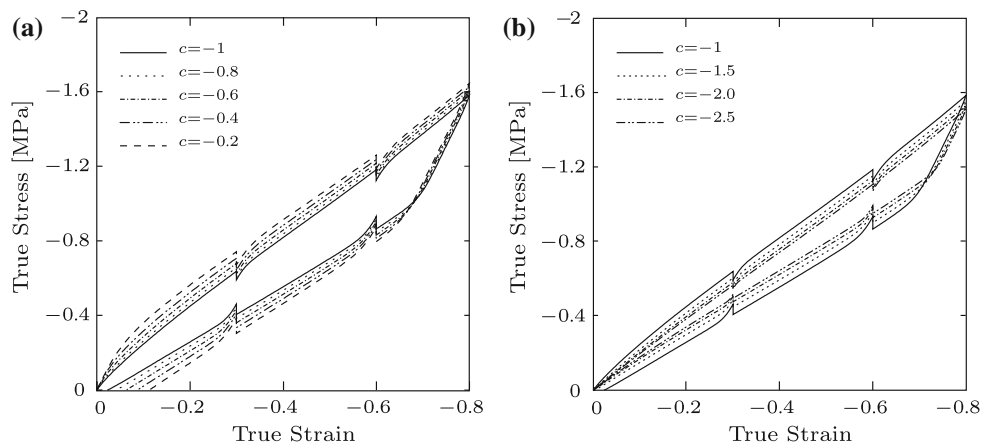
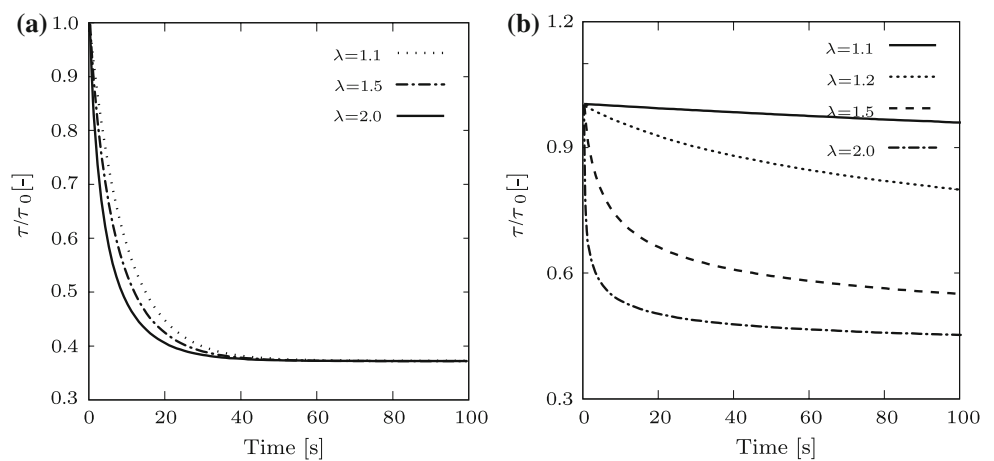


Fig. 6 Normalized relaxation curves: **a** Parameters $\lambda = 0$ and $m = 1$ yield a materially linear evolution law. **b** The evolution law is purely energy activated for parameters $\lambda = 0$ and $m = 4$



viscoelasticity models. The higher the stretch level, the steeper the descent of the curves as a consequence of growth in metrically induced nonlinearity in the evolution law. It is noteworthy to mention that the steep initial slope is associated with the singularity of the evolution law at $\dot{\gamma}_{chain} = 1$. Figure 7b depicts the stress relaxation curves of the model for $m = 4$ and $c = -1$ at different stretch levels.

Fig. 7 Normalized relaxation curves: **a** Parameters $\lambda = -1$ and $m = 1$ depict relaxation kinetics of entanglement. **b** The *BB* model with arbitrary parameters $\lambda = 4$ and $c = -1$

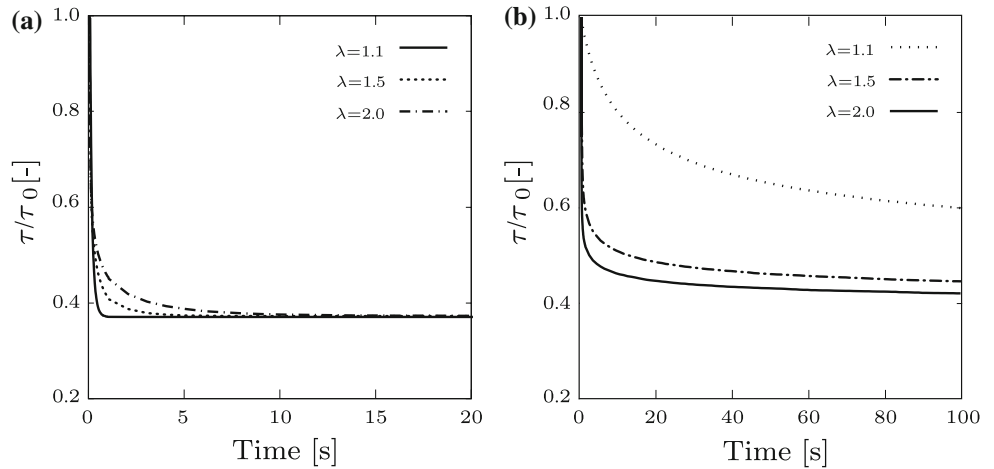
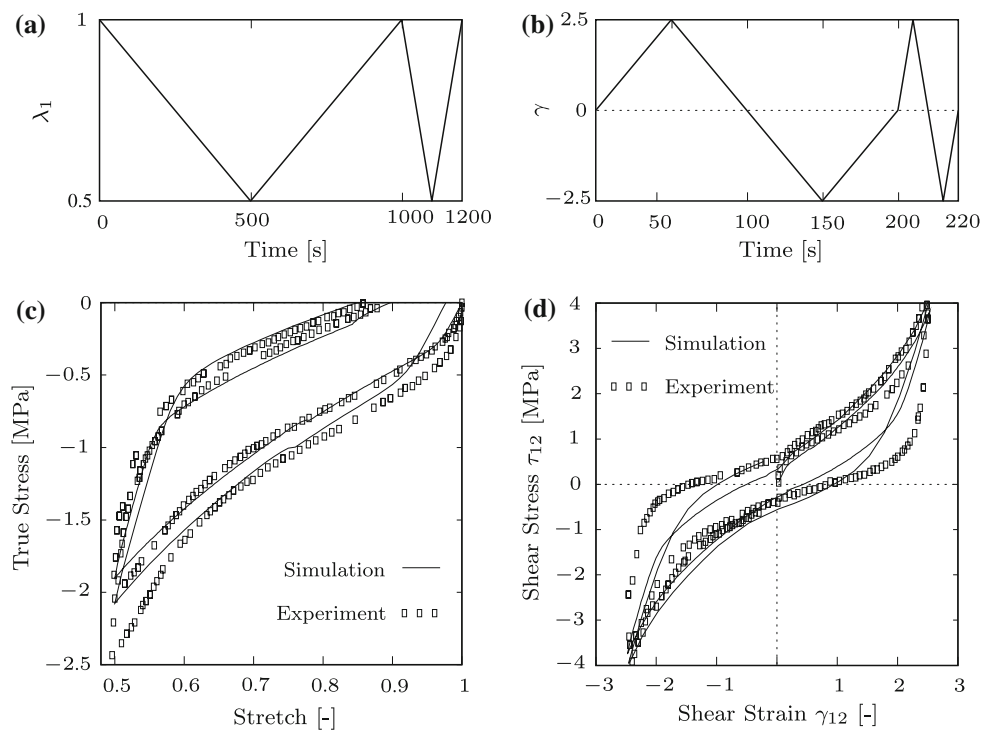


Fig. 8 Numerical simulation of cyclic **c** uniaxial compression, **d** simple shear tests of HDR at different strain rates. Experiments are taken from Amin et al. [25]. **a** and **b** shows the corresponding loading curves to **c** and **d**, respectively

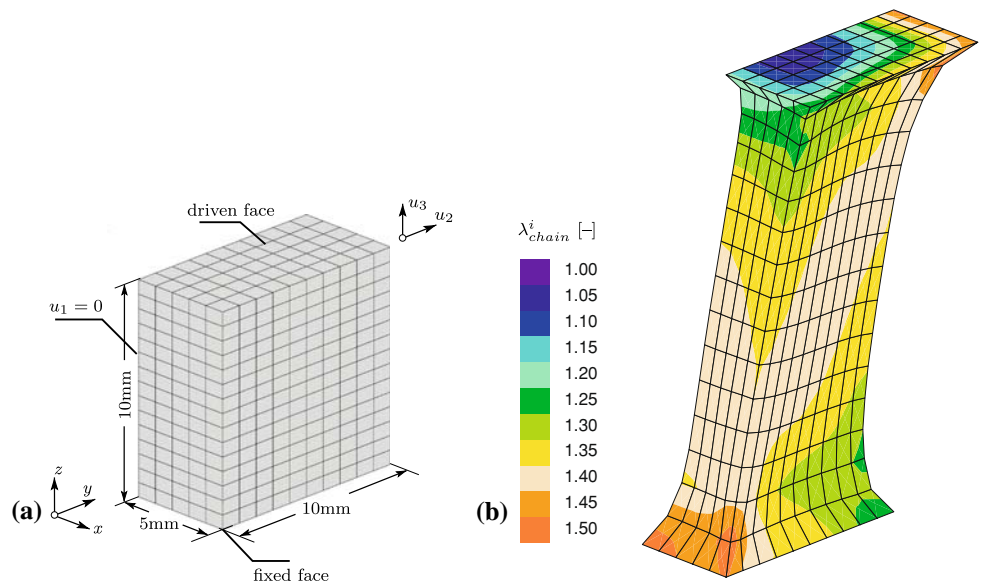


4.3 Example 3

In this example, the modeling capabilities of the *BB* model will be investigated with respect to recently published experimental data. Amin et al. [25] proposed a new evolution law for uniaxial compression and simple shear based on the finite viscoelasticity constitutive framework of Fig. 8c and d. The same material parameters are used for Huber and Tsakmakis [1]. They have reported different rate sensitivities for loading and unloading for high damping rubbers (HDR) and natural rubber (NR) under simple shear and uniaxial compression. Taking different rate sensitivities associated with loading and unloading into account, they proposed a nonlinear viscosity law $\eta = f(\tau : \mathbf{d}, \mathbf{b}, \text{dev} \bar{\tau}^v)$ to attach different viscosities to loading and unloading processes.

The Figures 30 and 31 of Amin et al. [25] are taken as a benchmark for comparison with the *BB* model. Loading curves are in better correlation than unloading curves. The perturbation parameter $\epsilon = 0.02$ is used in order to regularize the singularity around $\lambda_{chain} \approx 1$. Under uniaxial compression, the agreement of the *BB* model with the experiment is quite good (see Fig. 8c). For simple shear tests, loading curves are in better correlation than unloading curves.

Fig. 9 Finite element analysis of a cube driven in lateral and longitudinal directions: **a** Geometry, boundary conditions and FE discretization, **b** deformed shape at $t = 10$ s, contour plots depict the inelastic network stretch λ_{chain}^i



The parameter λ_{chain}^i improves for both compression and shear tests if the identification process takes into account a single mode of loading. In both uniaxial compression and simple shear tests, the *BB model* is in a better agreement with the experimental data than the model proposed by Amin et al. [25].

4.4 Example 4

In the following, we investigate the model response for a three-dimensional boundary value problem. During numerical simulation, different strain rates as well as nonhomogeneous deformation states are involved at each material point. Hence, it allows us to evaluate the overall response of the algorithmic setting proposed. The geometry and the boundary conditions of the problem are depicted in Fig. 9. All degrees of freedom are fixed at the bottom surface. At top surface, the specimen is driven with $u_2 = u_3 = u(t)$, whereas displacement in x -direction is held constant $u_1 = 0$. Due to symmetry, only half of the cube is discretized leading to $u_1 = 0$ at $x = 0$. The FE-mesh consists of 510×15 Q1/P0 brick elements. The specimen is monotonically loaded with a tip displacement rate $\dot{u}(t) = 10$ mm/s up to $t = 1$ s and kept stretched for relaxation. The incremental loading is performed at constant time step $\Delta t = 0.04$ s. The deformed configuration and the distribution of inelastic network stretch at $t = 10$ s is depicted in Fig. 9b. The material parameters used are identical to those of the first example. The FE-analysis shows that the numerical implementation is very efficient and quadratic convergence is observed which validates the performance of the proposed algorithmic setting.

4.5 Convergence and stability

As mentioned before, an explicit update of λ_{chain}^i destroys the unconditional stability of the backward Euler scheme adopted for the integration of the evolution law. The effect of this assumption on the global stability and convergence behaviour of the boundary value problem is investigated using the material parameters given in Sect. 4.3. A square cuboid with dimensions 25×20 mm (height = 20 mm) is subjected to uniaxial tension and macroscopic simple shear. Half of the cuboid is analysed due to symmetry. The FE-mesh consists of $10 \times 5 \times 10$ Q1/P0 brick elements. In the first set of experiments, the cuboid is subjected to a uniaxial tension up to $\lambda = 3.5$ with a loading rate of $\dot{\lambda} = 0.05$ /s. The specimen is investigated for time increments $\Delta t = 0.5$ s, $\Delta t = 1.0$ s, $\Delta t = 2.0$ s, $\Delta t = 4.0$ s, respectively. Stress-stretch curves for different time increments are depicted in Fig. 10. The simulations are repeated for $c = 0$ and $c = -0.8$. For $c = 0$, the material law does not include the chain relaxation kinetics and the algorithm proposed is implicitly updated totally. For $c = 0$, the simulation begins with an initial shear modulus $\mu + \mu_v$ (Fig. 10a) whereas for $c = -0.8$, due to initially zero viscosity, the shear modulus is equivalent to the shear modulus of the elastic branch (see Fig. 10b). The accuracy of the algorithm is satisfactory even at large time increments. The number of iterations per time step is shown in Fig. 10. The interesting outcome of the investigation is that there is no sign of deterioration in the convergence for $c = -0.8$. The above mentioned procedure is repeated for simple shear up to macroscopic shear strain $\gamma = 2.5$ and $\dot{\gamma} = 0.05$ /s (see Fig. 11). The simulation reveals non-homogeneous deformations satisfying the macroscopic simple shear deformation in

Fig. 10 Engineering stress versus stretch curves for **a** $c = 0$ and **b** $c = -0.8$, respectively. Solution for incremental time steps $\Delta t = 0.5\text{ s}$, $\Delta t = 1\text{ s}$, $\Delta t = 2\text{ s}$, $\Delta t = 4\text{ s}$ and a reference elastic solution $\Delta t = 0.5\text{ s}$ are given

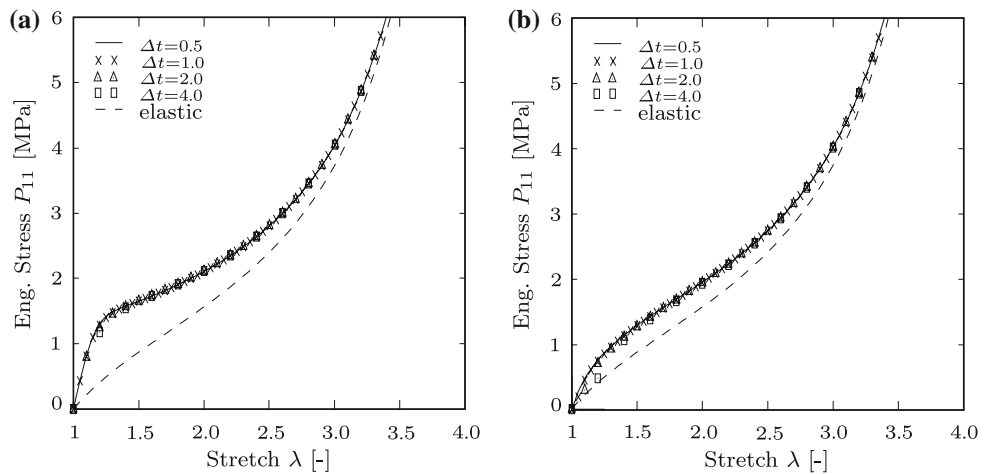
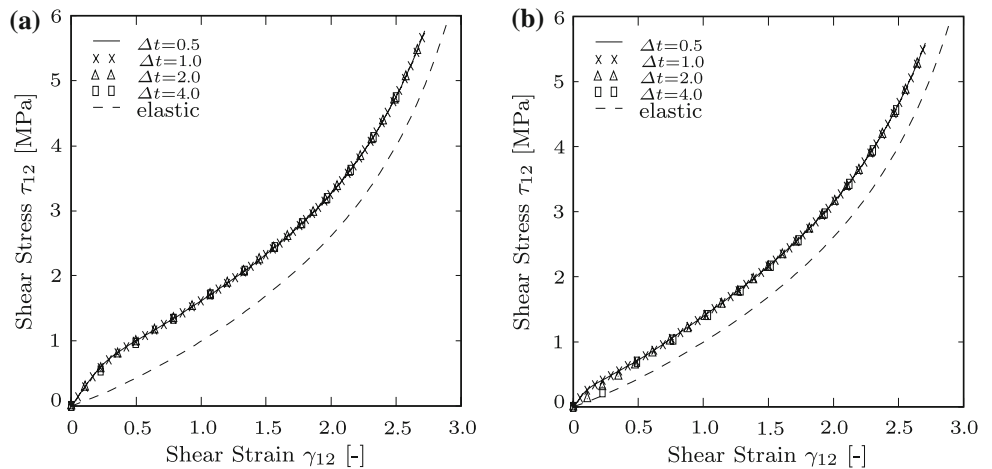


Fig. 11 Shear stress versus strain curves for **a** $c = 0$ and **b** $c = -0.8$, respectively. Solution for incremental time steps $\Delta t = 0.5\text{ s}$, $\Delta t = 1\text{ s}$, $\Delta t = 2\text{ s}$, $\Delta t = 4\text{ s}$ and a reference elastic solution $\Delta t = 0.5\text{ s}$ are given



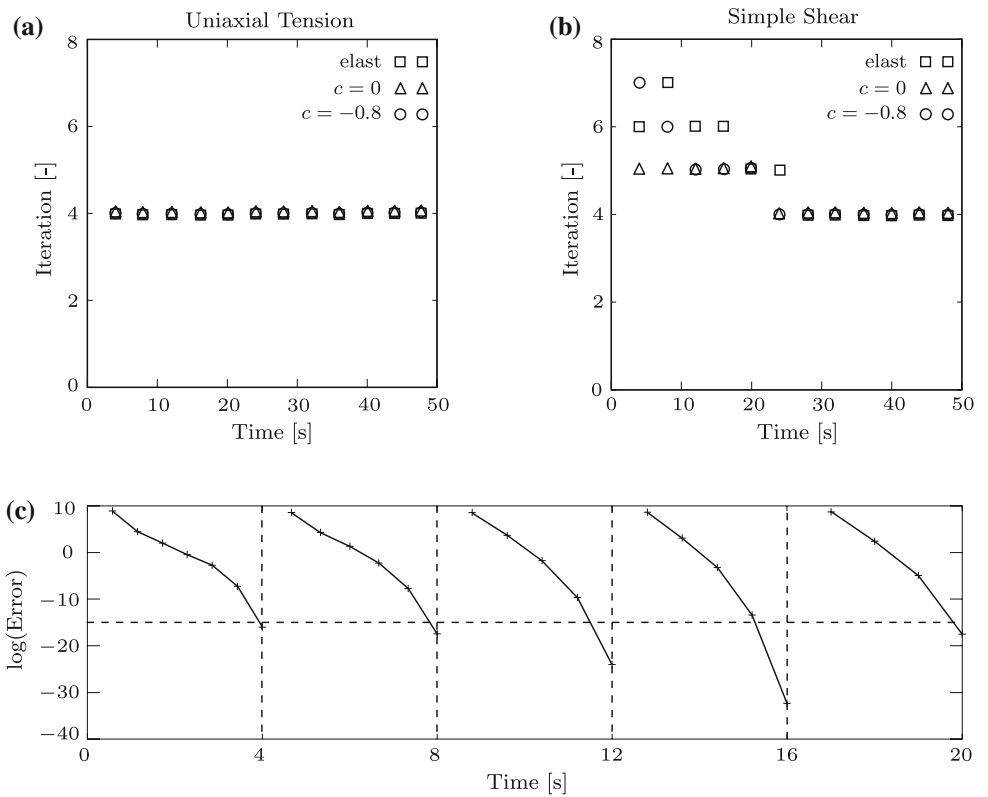
an average-sense as expected. Stress-stretch curves for $c = 0$ are related to the singularity at the beginning. The algorithm for $c = -0.8$ is very stable and does not terminate as long as the reference elastic solution is convergent. Introduced viscosity improves the convergence of the elastic reference solution.

Fig. 12b. Except for the first few iterations where the model includes a singularity at $\lambda_{\text{chain}}^i = 1$, the convergence is quadratic (Fig. 12c). The problem can be overcome by increasing parameter ϵ ($\epsilon = 0.001$ is taken throughout the simulations). This is related to the physics of the problem where the relaxation time approaches zero (see Fig. 7b). For accuracy, small time steps should be preferred at the beginning where $\lambda_{\text{chain}}^i \approx 1$.

Uniaxial tension simulations show quadratic convergence independent of the time increment. Under simple shear, the global FE solution is quadratically convergent for the reference elastic solution for both small and large time increments. The number of iterations is higher at the first few time steps for larger time increments. Viscoelastic simulations show quadratic convergence compared to the reference elastic solution. For the update of the inelastic metric, an elastic predictor-inelastic corrector algorithm is utilized. During inelastic corrector step, an implicit integration scheme is adopted. The resulting non-linear set of equations are solved iteratively by a Newton

5 Conclusion

Fig. 12 Number of iterations versus time for a uniaxial tension and a simple shear simulations in case of ground state elastic reference solution, parameter $c = 0$ and $c = -0.8$, respectively. Logarithmic plot of the error (energy norm), evenly distributed within the time increment $\Delta t = 4$ s for the simple shear simulation with $c = -0.8$



iteration. The consistent tangent moduli are computed from where a cautious intermediate stress expression obtained by the pull-back of the Kirchhoff stress to the intermediate configuration by the trial elastic deformation gradient. This scheme is very attractive since it circumvents the rate of the inelastic deformation gradient. Then, the formulation is successfully implemented into FEAP and validated with respect to uniaxial compression data of Bergström and Boyce. The parametric study has been carried out in order to assess the influence of material parameters on the overall behaviour. The same procedure is repeated for the parameters appearing in the evolution law and their influence on relaxation curves has been studied. The model is shown to fit well with recently published experimental data. The algorithm is validated with a benchmark boundary value problem. Finally, the convergence and stability of the proposed algorithm is elaborated which demonstrates the efficiency of the algorithmic setting.

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Appendix A: Volumetric-isochoric split of the deformation gradient and its rates

We begin with the split of F into spherical and unimodular part

$$F = \bar{F}F^{vol} \tag{A.1}$$

$$\bar{F} = J^{-1/3}F \text{ and } F^{vol} = J^{1/3}I. \tag{A.2}$$

Now, consider the spatial velocity gradient $d = \dot{F}F^{-1}$;

$$\begin{aligned} d &= \dot{F}F^{-1} \stackrel{(A.1)}{=} (\dot{\bar{F}}F^{vol} + \bar{F}\dot{F}^{vol})(F^{vol-1}\bar{F}^{-1}) \\ &= \dot{\bar{F}}\bar{F}^{-1} + \bar{F}\dot{F}^{vol}F^{vol-1}\bar{F}^{-1} \\ &\stackrel{(A.2)}{=} \dot{\bar{F}}\bar{F}^{-1} + \bar{F}(\frac{1}{3}J^{-2/3}\dot{J}J^{-1/3}I)\bar{F}^{-1} \\ &= \dot{\bar{F}}\bar{F}^{-1} + \frac{1}{3}\dot{J}J^{-1}I \\ &= \bar{d} + d^{vol}. \end{aligned} \tag{A.3}$$

We observe that
$$\bar{d} = \partial_{\bar{F}}J : \dot{\bar{F}} = JF^{-T} : \dot{\bar{F}} = J \text{tr}[\dot{\bar{F}}F^{-1}] = J \text{tr}[I]. \tag{A.4}$$

Inserting (A.4) into (A.3), we end up with an additive decomposition of the spatial velocity gradient into volumetric and unimodular contributions

$$\bar{d} = \dot{\bar{F}}\bar{F}^{-1} = I - \frac{1}{3}J \text{tr}[I]I = I^{iso}. \tag{A.5}$$

Furthermore,

$$\begin{aligned} d &:= \text{sym}(gI) = \text{sym}(gI^{iso} + gI^{vol}) \\ &= \text{sym}(gI^{iso}) + \text{sym}(gI^{vol}) \\ &= d^{iso} + d^{vol}. \end{aligned} \tag{A.6}$$

Therefore, we can rewrite the Clausius-Planck inequality in the form

$$\mathcal{D} = [\boldsymbol{\tau}_{\text{vol}} + \boldsymbol{\tau}_{\text{iso}}] : [\mathbf{d}^{\text{vol}} + \mathbf{d}^{\text{iso}}] - [\dot{U} + \dot{\Psi}] \geq 0. \quad (\text{A.7})$$

Examining that $\boldsymbol{\tau}_{\text{vol}} : \mathbf{d}^{\text{iso}} = 0$ and $\boldsymbol{\tau}_{\text{iso}} : \mathbf{d}^{\text{vol}} = 0$, (A.7) takes the following form

$$\mathcal{D} = \boldsymbol{\tau}_{\text{vol}} : \mathbf{d}^{\text{vol}} + \boldsymbol{\tau}_{\text{iso}} : \mathbf{d}^{\text{iso}} - [\dot{U} + \dot{\Psi}] \geq 0. \quad (\text{A.8})$$

We obtain a stronger constraint by decoupling the volumetric and the isochoric parts in the dissipation inequality, leading to

$$\mathcal{D}_{\text{vol}} = \boldsymbol{\tau}_{\text{vol}} : \mathbf{d}^{\text{vol}} - \dot{U} \geq 0 \quad (\text{A.9})$$

and

$$\mathcal{D}_{\text{iso}} = \boldsymbol{\tau}_{\text{iso}} : \mathbf{d}^{\text{iso}} - \dot{\Psi} \geq 0. \quad (\text{A.10})$$

Hence, one obtains a discrete representation of the Duhem inequality for purely isochoric processes of inelasticity.

Appendix B: Derivation of the local tangent for Newton iteration

In this Appendix, the derivation of local tangent \mathcal{K}_{ab} is shown. We recall the residual expression

$$r_a := \varepsilon_a + \frac{\Delta t}{\sqrt{2}} \frac{\dot{\gamma}}{\tau_v} \text{dev} \tau_a - \varepsilon_a^{\text{tr}} = 0. \quad (\text{B.1})$$

The local tangent is defined as

$$\mathcal{K}_{ab} := \left. \frac{\partial r_a}{\partial \varepsilon_b} \right|_{\varepsilon = \varepsilon^k} \quad (\text{B.2})$$

and can be derived from (B.1)

$$\mathcal{K}_{ab} = \delta_{ab} + \frac{\Delta t}{\sqrt{2}} \text{dev} \tau_a \frac{\partial (\dot{\gamma}/\tau_v)}{\partial \varepsilon_b} + \frac{\Delta t}{\sqrt{2}} \frac{\dot{\gamma}}{\tau_v} \frac{\partial \text{dev} \tau_a}{\partial \varepsilon_b}. \quad (\text{B.3})$$

Taking the derivatives in (B.3), one finally obtains the compact definition

$$\mathcal{K}_{ab} = \delta_{ab} + \beta_1 \text{dev} \tau_a \mathcal{D}_b + \beta_2 \bar{\mathcal{T}}_{ab}, \quad (\text{B.4})$$

along with the terms defined below:

$$\beta_1 = \frac{\Delta t}{2\sqrt{2}} (m-1) \frac{\dot{\gamma}_0}{\dot{\tau}^m} (\lambda_{\text{chain}}^i)^c \tau_v^{m-3}, \quad (\text{B.5})$$

$$\beta_2 = \frac{\Delta t}{\sqrt{2}} \frac{\dot{\gamma}}{\tau_v}, \quad (\text{B.6})$$

$$\mathcal{D}_b = \sum_{c=1}^3 \text{dev} \tau_c \mathcal{I}_{cb}, \quad (\text{B.7})$$

$$\bar{\mathcal{T}}_{ab} = \mathcal{I}_{ab} - \frac{1}{3} \sum_{c=1}^3 \mathcal{I}_{cb}, \quad (\text{B.8})$$

$$\begin{aligned} \mathcal{I}_{ab} &:= \frac{\partial \tau_a}{\partial \varepsilon_b} \\ &= \frac{2}{3} \mu_v \frac{3 - \lambda_1^{e2}}{1 - \lambda_r^{e2}} \lambda_a \lambda_b \delta_{ab} + \frac{4}{9} \frac{\mu_v}{N_v} \frac{1}{(1 - \lambda_r^{e2})^2} \lambda_a^2 \lambda_b^2. \end{aligned} \quad (\text{B.9})$$

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